A finite strain kinematic hardening constitutive model based on
Hencky strain: General framework, solution algorithm and application
to shape memory alloys

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A B S T R A C T
The logarithmic or Hencky strain measure is a favored measure of strain due to its remarkable properties in large deformation problems. Compared with other strain measures, e.g., the commonly used Green–Lagrange measure, logarithmic strain is a more physical measure of strain. In this paper, we present a Hencky-based phenomenological finite strain kinematic hardening, non-associated constitutive model, developed within the framework of irreversible thermodynamics with internal variables. The derivation is based on the multiplicative decomposition of the deformation gradient into elastic and inelastic parts, and on the use of the isotropic property of the Helmholtz strain energy function. We also use the fact that the corotational rate of the Eulerian Hencky strain associated with the so-called logarithmic spin is equal to the strain rate tensor (symmetric part of the velocity gradient tensor). Satisfying the second law of thermodynamics in the Clausius–Duhem inequality form, we derive a thermodynamically-consistent constitutive model in a Lagrangian form. In comparison with the available finite strain models in which the unsymmetric Mandel stress appears in the equations, the proposed constitutive model includes only symmetric variables. Introducing a logarithmic mapping, we also present an appropriate form of the proposed constitutive equations in the time-discrete frame. We then apply the developed constitutive model to shape memory alloys and propose a well-defined, non-singular definition for model variables. In addition, we present a nucleation-completion condition in constructing the solution algorithm. We finally solve several boundary value problems to demonstrate the proposed model features as well as the numerical counterpart capabilities.

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1. Introduction

In recent years much attention has been devoted to the development of constitutive laws for large deformations of solids. The logarithmic strain, sometimes referred to as true or natural strain, introduced by Hencky (1928) and also called Hencky
strain, is a favored measure of strain due to its remarkable properties at large deformations in solid mechanics, materials science and metallurgy. A remarkable property is that among all finite strain measures, only the spherical and the deviatoric parts of the Hencky strain can, in an additive manner, separate the volumetric deformation and the isochoric deformation from the total deformation, whereas the just-mentioned two deformation modes are inseparably coupled with each other in either of the spherical part and the deviatoric part of any other strain measure (Xiao et al., 2004).

Moreover, the logarithmic strain possesses certain intrinsic far-reaching properties that establish its favored position in all possible strain measures. One reason of such properties was clarified by disclosing an important relation, i.e., the Eulerian logarithmic strain is the unique strain measure that its corotational rate (associated with the so-called logarithmic spin) is the strain rate tensor. In other words, the strain rate tensor, \( d \), is the corotational rate of the Hencky strain tensor associated with the logarithmic spin tensor. Such a result has been introduced by Reinhardt and Dubey (1995) as D-rate and by Xiao et al. (1997) as log-rate. The work-conjugate pair of Eulerian Hencky strain and Cauchy stress, being natural or true strain and stress measures, respectively, are of most interest in constitutive modeling beyond the small strain regime (Xiao et al., 2006).

For its considerable advantages, logarithmic strain has been used in constitutive modeling of solids by many authors (see, e.g., Xiao et al., 2000, 2001; Bruhns et al., 1999; Yeganeh and Naghdabadi, 2006; Naghdabadi et al., 2005; Peric et al., 1992; Reinhardt and Dubey, 1996; Criscione et al., 2000; Miehe et al., 2002; Lin and Schomburg, 2003; Xiao and Chen, 2002; Müller and Bruhns, 2006).

There has also been a lot of effort in the literature using Green–Lagrange strain and the second Piola–Kirchhoff stress which are finite strain and stress as well as work-conjugate measures in nonlinear solid mechanics. However, the physical meaning of Green–Lagrange strain is not precise and direct unless the magnitude is small. Sometimes Green–Lagrange strain should take a very large positive value. (2) strain should go to +\( \infty \) when stretch goes to +\( \infty \). This means that under a very large stretching condition, strain should take a very large positive value. (2) strain should go to –\( \infty \) when stretch goes to zero. This means that under a high compression condition, strain should take a very large negative value. As discussed by Darijani and Naghdabadi, the well known Seth–Hill strain measures do not satisfy the both requirements simultaneously.\(^2\) We emphasis that the Hencky strain, belonging to both Seth–Hill and Darijani–Naghdabadi measures, is the only member of Seth–Hill class that satisfies both requirements.

There are still more motivations to develop a Hencky-strain based constitutive model. For example, for a Hookean-type constitutive relation, only the Hencky-based one is useful at moderately large elastic stretches (Anand, 1979, 1986). Another motivation is related to the physical kinematical constraints, for instance, in shape memory alloys (SMAs), to capture the transformation-induced strain saturation, it is necessary to define a kinematical constraint on the inelastic true strain.

This constraint has so far been enforced through the Green–Lagrange strain norm. However, the use of Green–Lagrange strain is reasonable for small strain, large rotation deformations. Moreover, the incompressibility constraint takes a very simple form when Hencky strain is used (Xiao et al., 2004).

In this study, based on the Hencky strain, we develop a finite strain constitutive model. To this end, the paper is organized as follows. Section 2 presents some preliminaries. In Section 3, based on a multiplicative decomposition of the deformation gradient into elastic and inelastic parts, we present a finite-strain non-associative constitutive model. In Section 4, we introduce a logarithmic mapping to propose the time-discrete constitutive model. In Section 5, we apply the developed constitutive model to shape memory alloys and solve several boundary value problems in Section 6. We finally draw conclusions in Section 7.

### 2. Preliminaries

According to the polar decomposition theorem, the deformation gradient \( F \) with \( J = \det(F) > 0 \) is uniquely decomposed as

\[
F = RU = VR, \tag{1}
\]

where \( R \) is a proper orthogonal tensor and \( U \) and \( V \) are symmetric positive definite right and left stretch tensors, respectively.

A multiplicative decomposition of \( F \) into dilatational and distortional parts is defined as

\[
F = \left( J^{1/3} \right) \bar{F} = J^{1/3} \bar{F}, \tag{2}
\]

where \( \mathbf{1} \) is the second-order identity tensor, and \( \det(\bar{F}) = 1 \). The terms \( J^{1/3} \mathbf{1} \) and \( \bar{F} \) are associated, respectively, with volume-changing and volume-preserving deformations of the material (Holzapfel, 2000).

Then, the right and left Cauchy–Green deformation tensors are, respectively, defined as

\[
\mathbf{C} = \bar{F}\bar{F}^T = U^2 = J^{2/3} \bar{C} = J^{2/3} \bar{U}^2, \quad \mathbf{b} = FF^T = V^2 = J^{2/3} \bar{b} = J^{2/3} \bar{V}^2, \tag{3}
\]

where \( \det(\bar{C}) = \det(\bar{U}) = 1 \) and \( \det(\bar{b}) = \det(\bar{V}) = 1 \).

\(^2\) For example, Green–Lagrange strain measure, \( E = \frac{1}{2} \left( U^2 - 1 \right) \rightarrow -\frac{1}{2} \mathbf{1} \) when \( U = \mathbf{0} \).
The material and spatial Hencky (or logarithmic) strain tensors \(H\) and \(h\) read, respectively, as (Lubarda, 2002)\(^3\):

\[
H = \log U = \frac{1}{2} \log C = \bar{H} + \frac{1}{3} \theta \mathbf{1},
\]

\[
h = \log V = \frac{1}{2} \log b = \bar{h} + \frac{1}{3} \theta \mathbf{1},
\]

where

\[
\theta = \log(f), \quad \bar{H} = \log U = \frac{1}{2} \log \bar{C}, \quad \bar{h} = \log V = \frac{1}{2} \log \bar{b},
\]

Moreover, the velocity gradient tensor \(l\) is given as

\[
l = FF^{-1} = \bar{l} + \frac{1}{3} \theta \mathbf{1},
\]

where

\[
\bar{l} = FF^{-1},
\]

The symmetric and skew symmetric parts of \(l\) supply, respectively, the strain rate tensor \(d\) and the vorticity tensor \(w\), i.e.

\[
d = \text{sym}(l) = \bar{d} + \frac{1}{3} \theta \mathbf{1},
\]

\[
w = \text{skew}(l) = \bar{w},
\]

where \(\bar{d} = \text{sym}(\bar{l})\) and \(\bar{w} = \text{skew}(\bar{l})\), while \(\text{sym}(A) = (1/2)(A + A^T)\) and \(\text{skew}(A) = (1/2)(A - A^T)\) compute the symmetric and skew symmetric parts of an arbitrary tensor \(A\), respectively.

Taking the time derivative of relation \(C = FF^T\) and using (6)\(^1\) and (8)\(^1\), it can be shown that:

\[
\dot{C} = 2F^T dF.
\]

Corotational rate of an Eulerian tensor \(\mathcal{A}\) associated with the rotating frame having spin \(\Omega\) is defined as

\[
\dot{\mathcal{A}} = \dot{\mathcal{A}} - \Omega \mathcal{A} + \mathcal{A} \Omega,
\]

where \(\dot{\mathcal{A}}\) is the material time derivative of \(\mathcal{A}\) associated with the fixed frame and \(\dot{\mathcal{A}}\) is the corotational rate of \(\mathcal{A}\) (Xiao et al., 1997; Ghavam and Naghdabadi, 2007).

We now double contract both sides of (10) with an arbitrary second-order symmetric tensor \(B\); assuming \(\mathcal{A}\) symmetric and after some mathematical manipulation, we obtain:

\[
\mathcal{A} : B = \dot{\mathcal{A}} : B + 2 \Omega : (\mathcal{A}B),
\]

where: indicates double contraction, defined as \(A : B = A_{ij}B_{ij}\) for second-order tensors.

If \(\mathcal{A}\) and \(B\) are coaxial (besides being symmetric), \(\mathcal{A}B\) is also symmetric and its double contraction with the skew-symmetric tensor \(\Omega\) vanishes. Therefore, we conclude the following identity for any arbitrary spin tensor \(\Omega\):

\[
\text{if } \mathcal{A} \text{ and } B \text{ are symmetric and coaxial } \Rightarrow \dot{\mathcal{A}} : B = \dot{\mathcal{A}} : B.
\]

The corotational rate of the Eulerian (or spatial) Hencky strain \(h\) associated with the so-called logarithmic spin\(^4\) \(\Omega^{\log}\) is identical to the Eulerian stretching \(d\), and \(h\) is the only strain enjoying this property (Xiao et al., 1997). According to (10), this means:

\[
h^{\log} = \dot{h} - \Omega^{\log} h + h \Omega^{\log} = \dot{d}.
\]

3. A kinematic hardening constitutive model based on Hencky strain

In this section, we use the multiplicative decomposition of the deformation gradient into elastic and inelastic parts and, defining Helmholtz strain energy function in terms of elastic and inelastic Hencky strains, we develop a thermodynamically-consistent finite strain kinematic hardening constitutive model.

\(^3\) The material and spatial Hencky strains are computed through spectral decomposition, i.e.,

\[
H = \sum_{i=1}^{3} \log(\lambda_i) \mathbf{e}_i \otimes \mathbf{e}_i \quad \text{and} \quad h = \sum_{i=1}^{3} \log(\lambda_i) \bar{\mathbf{e}}_i \otimes \bar{\mathbf{e}}_i,
\]

where \(\lambda_i\) are the three principal stretches (possibly repeated), while \(\mathbf{e}_i\) and \(\bar{\mathbf{e}}_i\) are the corresponding principal directions of the right and left stretch tensors, respectively.

\(^4\) In this study, only the existence of such spin is utilized and the explicit form of \(\Omega^{\log}\) is not needed; however we refer to e.g., Reinhardt and Dubey (1995), Xiao et al. (1997), Ghavam and Naghdabadi (2007) for more details.
3.1. Constitutive model development

Following a well-established approach adopted in plasticity (Lubarda, 2002; Haupt, 2002) and already used for SMAs (see, e.g., Helm, 2001; Auricchio, 2001; Ziolkowski, 2007; Reese and Christ, 2008; Christ and Reese, 2009; Arghavani et al., 2010d; Thamburaja, 2010 among others), we assume a multiplicative decomposition of the deformation gradient into an elastic part \( F^e \), defined with respect to an intermediate configuration, and an inelastic one \( F^\text{in} \), defined with respect to the reference configuration,\(^5\) i.e.

\[
F = F^e F^\text{in}.
\]

Moreover, we utilize the deformation gradient decomposition into volumetric and distortional parts, as follows:

\[
F^e = f^e F,
\]

\[
F^\text{in} = j^\text{in} F^\text{in},
\]

where \( f^e = \det (F^e) \), \( j^\text{in} = \det (F^\text{in}) \) and \( \det (F) = \det (F^\text{in}) = 1 \).

Similarly, we may define the following tensors:

\[
C^e = F^e F^{e\text{T}}, \quad C^\text{in} = F^\text{in} F^{\text{inT}}, \quad b^e = F^e F^e, \quad b^\text{in} = F^\text{in} F^\text{in},
\]

\[
C^e = F^e F^{e\text{T}}, \quad C^\text{in} = F^\text{in} F^{\text{inT}}, \quad b^e = F^e F^e, \quad b^\text{in} = F^\text{in} F^\text{in},
\]

\[
U^e = C^{e1/2}, \quad U^\text{in} = C^{\text{in1/2}}, \quad V^e = b^{e1/2}, \quad V^\text{in} = b^{\text{in1/2}},
\]

\[
H^e = \log U^e, \quad H^\text{in} = \log U^\text{in}.
\]

Combining (14) and (15), we obtain:

\[
F = (f^e j^\text{in})^{-1} F^e F^\text{in} = f^e F,
\]

where

\[
j = f^e j^\text{in} \quad \text{and} \quad F = F^e F^\text{in}.
\]

Substituting (18) into (8), we obtain:

\[
d = FF^{-1} = d^e + \text{sym} \left( F^e F^{e\text{T}} \right) \quad \text{and} \quad \theta = \theta^e + \theta^\text{in},
\]

where, \( d^e = \text{sym} \left( F^e F^{e\text{T}} \right) \), \( l^e = F^e F^{e\text{T}}^{-1} \), \( \theta^e = \log (f^e) \) and \( \theta^\text{in} = \log (j^\text{in}) \).

In order to satisfy the principle of material objectivity, the Helmholtz free energy should depend on \( F^e \) only through the elastic right stretch tensor; it is moreover assumed to be a function of \( F^\text{in} \) through the inelastic right stretch tensor and of the temperature.\(^6\) \( T \). Finally, we express the Helmholtz free energy per unit undeformed volume, decomposed additively in the following form:

\[
\psi = \psi \left( \theta^e, \theta^\text{in}, H^e, H^\text{in}, T \right) = \phi^e (\theta^e) + \psi^e (H^e) + \phi^\text{in} (\theta^\text{in}) + \psi^\text{in} (H^\text{in}, T) = W^e (H^e) + W^\text{in} (H^\text{in}, T),
\]

where \( W^e (H^e) \) is a hyperelastic strain energy function decomposed into \( \phi^e (\theta^e) \) and \( \psi^e (H^e) \), representing the volumetric and distortional elastic strain energies due to the elastic material deformations, respectively. Similarly, the term \( W^\text{in} (H^\text{in}, T) \) represents the additional amount of stored energy due to inelastic hardening (e.g., phase transformation, plastic hardening and so on), decomposed into volumetric and isochoric parts \( \phi^\text{in} (\theta^\text{in}) \) and \( \psi^\text{in} (H^\text{in}) \), respectively. For simplicity we do not consider the isotropic hardening effect, knowing that it can be easily included in the proposed constitutive model.

In this study, we assume \( \psi^e (H^e) \) and \( \psi^\text{in} (H^\text{in}, T) \) to be isotropic functions of \( H^e \) and \( H^\text{in} \), respectively.\(^7\) Therefore, \( \psi^e \) and \( \psi^\text{in} \) can be expressed as functions of their argument invariants. Since, the Lagrangian and Eulerian Hencky strains have the same invariants, we can express the Helmholtz free energy as a function of the Eulerian Hencky strains, i.e.

\[
\psi \left( \theta^e, \theta^\text{in}, H^e, H^\text{in}, T \right) = \psi \left( \theta^e, \theta^\text{in}, \tilde{h}^e, \tilde{h}^\text{in}, T \right) = \phi^e (\theta^e) + \psi^e (\tilde{h}^e) + \phi^\text{in} (\theta^\text{in}) + \psi^\text{in} (\tilde{h}^\text{in}, T),
\]

We now use Clausius–Duhem inequality form of the second law of thermodynamics:

\[
\dot{\sigma} : d - \langle \Psi + \eta \tilde{T} \rangle = \tau : d - \langle \Psi + \eta \tilde{T} \rangle \geq 0,
\]

\(^5\) We use inelastic terminology which is more general. For example when talking about plasticity, we can use \( F^p = F^e \) and when we are interested in SMA modeling, the inelasticity is due to phase transformation and we can use \( F^p = F^e \).

\(^6\) We remark that we do not consider a fully thermo-mechanical coupled model. In this way, the Helmholtz free energy should be more properly referred to as a temperature-parameterized function.

\(^7\) We refer to Henann and Anand (2009) and Gurtin and Anand (2005) for more details on objectivity as well as isotropy requirements.
where \( \eta \) is the entropy and \( \tau = \sigma \) is the Kirchhoff stress tensor.

Moreover, we make use of (8), and the additive decomposition of the Kirchhoff stress tensor \( \tau \) into a hydrostatic part \( \rho \mathbf{I} \) and a deviatoric part: \( \mathbf{S} = \tau - \rho \mathbf{I} \), where \( \rho = \text{tr} (\tau) / 3 \), and rewrite (22) as

\[
p \dot{\mathbf{S}} + \mathbf{S} : \dot{\mathbf{d}} - (\Psi + \eta \mathbf{I}) \geq 0.
\]  
(23)

We now substitute (19) and (21) into (23) to obtain:

\[
p \left( \dot{\mathbf{S}} + \frac{\partial \psi}{\partial \mathbf{h}} \right) + \mathbf{S} \left[ \dot{\mathbf{d}} + \text{sym} (\mathbf{F}^{-1} \mathbf{F}') \right] - \left( \frac{\partial \psi}{\partial \mathbf{h}} \dot{\mathbf{h}} + \frac{\partial \psi}{\partial \mathbf{l}} \dot{\mathbf{l}} + \frac{\partial \psi}{\partial \mathbf{w}} \dot{\mathbf{w}} - \eta \mathbf{I} \right) \geq 0.
\]  
(24)

As a consequence of isotropy, the symmetric tensors \( \mathbf{h}' \) and \( \partial \psi / \partial \mathbf{h}' \) as well as the symmetric tensors \( \mathbf{h}'' \) and \( \partial \psi / \partial \mathbf{h}'' \) are coaxial. According to (12), we can replace the time derivative of elastic and inelastic Hencky strains appearing in (24) with any corotational rate in the form of (10). We particularly make use of the corresponding logarithmic spin tensors, i.e., \( \Omega^{log-e} \) and \( \Omega^{log-in} \), which according to (13) result in:

\[
\frac{\partial \psi}{\partial \mathbf{h}} : \dot{\mathbf{h}} = \frac{\partial \psi}{\partial \mathbf{h}} : \dot{\mathbf{h}}^{log-e} = \frac{\partial \psi}{\partial \mathbf{h}} : \dot{\mathbf{d}},
\]  
(25)

and

\[
\frac{\partial \psi}{\partial \mathbf{h}} : \dot{\mathbf{h}}^{in} = \frac{\partial \psi}{\partial \mathbf{h}} : \dot{\mathbf{h}}^{log-in} = \frac{\partial \psi}{\partial \mathbf{h}} : \dot{\mathbf{d}}^{in},
\]  
(26)

where superscripts “log-e” and “log-in” represent the corotational rate associated with the logarithmic spins during the elastic and inelastic deformations, \( \mathbf{F}^{c} \) and \( \mathbf{F}^{n} \), respectively.

Substituting (25) and (26) into (24), after some mathematical manipulations, we obtain:

\[
\left( p - \frac{\partial \psi}{\partial \mathbf{h}} \right) \dot{\mathbf{h}} + \left( \mathbf{S} - \frac{\partial \psi}{\partial \mathbf{h}} \right) : \dot{\mathbf{d}} + \left( p - \frac{\partial \psi}{\partial \mathbf{h}} \right) \dot{\mathbf{l}}^{in} + \left( \mathbf{F}^{c} : \mathbf{F}^{n} \right) \mathbf{F}^{n} - \left( \eta + \frac{\partial \psi}{\partial \mathbf{h}} \right) \mathbf{T} \geq 0.
\]  
(27)

In deriving (27) we have used the symmetric property of both \( \mathbf{S} \) and \( \partial \psi / \partial \mathbf{h}^{in} \) implying, respectively:

\[
\mathbf{S} : \text{sym} (\mathbf{F}^{-1} \mathbf{F}') = \mathbf{S} : \left( \mathbf{F}^{-1} \mathbf{F}^{n} \right) = \mathbf{S} : \left( \mathbf{F}^{c} : \mathbf{F}^{n} \right),
\]  
(28)

and

\[
\frac{\partial \psi}{\partial \mathbf{h}} : \dot{\mathbf{d}}^{in} = \frac{\partial \psi}{\partial \mathbf{h}} : \dot{\mathbf{l}}^{in}.
\]  
(29)

The inequality (27) must be fulfilled for arbitrary thermodynamic processes, i.e. for arbitrary \( \dot{\mathbf{h}}, \dot{\mathbf{d}}, \dot{\mathbf{l}}^{in}, \dot{\mathbf{d}}^{in} \) and \( \mathbf{T} \). For arbitrary choices of the variables \( \dot{\mathbf{h}}, \dot{\mathbf{d}} \) and \( \mathbf{T} \), we may conclude\(^8\) (Haupt, 2002; Ottosen and Ristinmaa, 2005):

\[
p = \frac{\partial \psi}{\partial \mathbf{h}}, \quad \mathbf{S} = \frac{\partial \psi}{\partial \mathbf{h}}, \quad \eta = - \frac{\partial \psi}{\partial \mathbf{h}},
\]  
(30)

and the following simplified Clausius–Duhem inequality to guarantee the non-negativeness of the internal dissipation (required by the second law of thermodynamics):

\[
\left( p - \frac{\partial \psi}{\partial \mathbf{h}} \right) \dot{\mathbf{l}}^{in} + \left( \mathbf{F}^{c} : \mathbf{F}^{n} \right) \mathbf{F}^{n} \geq 0.
\]  
(31)

\(^8\) According to the work by Xiao et al. (1997), the logarithmic spin \( \Omega^{log-e} \) is introduced as

\[
\Omega^{log-e} (\mathbf{F}) = \mathbf{w} + \sum_{i,j=1}^{n} \left[ \frac{1}{1 - (b_i/b_j)} - \frac{2}{\text{log}(b_i/b_j)} \right] \mathbf{p}_i \mathbf{d}_i \mathbf{p}_j,
\]

where \( b_i = \lambda_i^2 \) are the eigenvalues of the left Cauchy–Green deformation tensor and \( \mathbf{P} \), are the eigenprojections subordinate to eigenvalues \( \lambda_i > 0 \). According to this relation, knowing the deformation gradient \( \mathbf{F}(t) \), the logarithmic spin \( \Omega^{log-e}(\mathbf{F}) \) can be computed. We now consider the multiplicative decomposition of the deformation gradient and introduce the logarithmic spins \( \Omega^{log-e} \) and \( \Omega^{log-in} \) as follows:

\[
\Omega^{log-e} = \Omega^{log-e}(\mathbf{F}) \quad \text{and} \quad \Omega^{log-in} = \Omega^{log-in}(\mathbf{F}).
\]

\(^9\) In the work by Xiao et al. (2000), they use the hypoelastic constitutive model \( \mathbf{d}^{e} = \mathbf{d}^{h} + \mathbf{d}^{\pi} \), where \( \mathbf{d}^{e} \) is a complementary hyperelastic potential and \( \pi = \mathbf{R}_{\pi} \mathbf{d}^{e} \) is the Eulerian stress while \( \mathbf{H} \) is the work-conjugate stress measure of the Lagrangean logarithmic strain \( \mathbf{h} \). However, the corresponding hypoelastic constitutive model of this work, when written in the framework of Xiao et al. (2000) work, takes the form \( \mathbf{d}^{e} = \mathbf{h}^{h} \mathbf{h}^{\pi} + \mathbf{h}^{\pi} \mathbf{h}^{h} \) which can be integrated to yield \( \mathbf{h}^{e} = \mathbf{h}^{h} / \mathbf{h}^{\pi} \). This shows that the hypoelastic form used in this work can be reduced to a hyperelastic constitutive model as observed in (30)_{1,2}. 
Considering (30), and the isotropic property of $\psi^e$, it can be shown that $\mathbf{F}^e \mathbf{s}^{Fe^e}$ is a symmetric tensor (see Eq. (118) in the Appendix A). Therefore, Eq. (31) reads as

$$q \delta^m + \lambda : \mathbf{d}^m \geq 0,$$

where the scalar and tensorial relative stresses, $q$ and $\lambda$, are defined as

$$q = p - \frac{\partial \phi^m}{\partial \mathbf{d}^m} = \frac{\partial \phi^e}{\partial \mathbf{e}^e} - \frac{\partial \phi^m}{\partial \mathbf{d}^m},$$

$$\lambda = m - x,$$

with

$$m = \mathbf{F}^e \mathbf{s}^{Fe^e}, \quad x = \frac{\partial \phi^m}{\partial \mathbf{e}^m}. \quad (34)$$

To satisfy the second law of thermodynamics (32), following Ottosen and Ristinmaa (2005), we introduce the convex potential $G(\lambda, q)$ and define the following evolution equations:

$$\mathbf{d}^m = \lambda \frac{\partial G(\lambda, q)}{\partial \lambda} \quad \text{and} \quad \dot{\mathbf{d}}^m = \lambda \frac{\partial G(\lambda, q)}{\partial q}, \quad (35)$$

where $\lambda$ is the consistency parameter.

Moreover, we consider a yield or limit function as follows:

$$F = f(\lambda, q) - R. \quad (36)$$

where the material parameter $R$ is the elastic region radius.

Similarly to plasticity, the consistency parameter and the limit function (or yield function) satisfy the Kuhn–Tucker conditions:

$$F \leq 0, \quad \lambda \geq 0, \quad \lambda F = 0. \quad (37)$$

Accordingly, the finite strain, non-associative constitutive model can be summarized as

- **Stress-like quantities:**

  $$p = \frac{\partial \phi^e}{\partial \mathbf{e}^e}, \quad q = \frac{\partial \phi^e}{\partial \mathbf{d}^e} - \frac{\partial \phi^m}{\partial \mathbf{d}^m}, \quad \mathbf{s} = \frac{\partial \psi^e}{\partial \mathbf{e}^e}, \quad \mathbf{m} = \mathbf{F}^e \mathbf{s}^{Fe^e}, \quad x = \frac{\partial \phi^m}{\partial \mathbf{e}^m}, \quad \lambda = m - x.$$

- **Limit function:**

  $$F = f(\lambda, q) - R.$$

- **Potential function:**

  $$G = G(\lambda, q).$$

- **Evolution equations:**

  $$\dot{\mathbf{d}}^m = \lambda \frac{\partial G(\lambda, q)}{\partial \lambda} \quad \text{and} \quad \dot{\mathbf{d}}^m = \lambda \frac{\partial G(\lambda, q)}{\partial q}.$$

- **Kuhn–Tucker conditions:**

  $$F \leq 0, \quad \lambda \geq 0, \quad \lambda F = 0.$$

### 3.2. Representation with respect to the reference configuration

In the previous section, using a multiplicative decomposition of the deformation gradient into elastic and inelastic parts, we derived a finite-strain constitutive model. But while $\mathbf{F}^e$ and $\mathbf{F}^m$ are related to the reference configuration, $\mathbf{F}^e$ is defined with respect to an intermediate configuration. Accordingly, tensor $\mathbf{s}$ has been defined with respect to the current configuration, while tensors $\mathbf{m}, \mathbf{x}, \lambda$ and $\mathbf{d}^m$ have been defined with respect to the intermediate configuration. Thus, we recast all equations in terms of quantities defined with respect to the reference configuration, allowing to express all equations in a Lagrangian form.

As already stated, $\psi^e$ depends on $\mathbf{h}^e$ only through its invariants which are the same as those of $\mathbf{H}^e$. Since $\mathbf{H}^e = 1/2 \log(\mathbf{C}^e)$, it is clear that the invariants of $\mathbf{H}^e$ can be expressed in terms of the invariants of $\mathbf{C}^e$ which are equal to those of $\mathbf{U}^m \mathbf{C} \mathbf{U}^m^t$. In fact, for the first invariant, the following identity holds:
where the isotropic property of \( \psi^e \) and according to the representation theorem,\(^{10} \) we now write:

\[
\frac{\partial \psi^e}{\partial \mathcal{H}} = \alpha_1 \mathbf{1} + 2 \alpha_2 \mathbf{h}^e + 4 \alpha_3 \mathbf{h}^2,
\]

where

\[
\alpha_i = \alpha_i(I_{1e}, II_{1e}, III_{1e}).
\]

It can be shown that (see Eq. (121) in the Appendix A):

\[
\mathbf{h}^e = \frac{1}{2} \mathbf{F}^{-1} \log(\mathbf{C}^{\text{inv}}) \mathbf{F}^T.
\]

We now substitute (42) into (40) and the result into (30), which leads to:

\[
s = \mathbf{F}^{-1} \mathbf{M} \mathbf{F}^T,
\]

where we have defined \( \mathbf{M} = \alpha_1 \mathbf{1} + \alpha_2 \log(\mathbf{C}^{\text{inv}}) + \alpha_3 (\log(\mathbf{C}^{\text{inv}}))^2 \).

We now pull back the quantities \( \mathbf{m} \) and \( \mathbf{x} \) (which have been expressed with respect to the intermediate configuration) to the reference configuration. To this end, we use the identity \( \mathbf{F}^{T} \mathbf{h}^{e} \mathbf{F}^{-T} = \mathbf{h}^{e} \) (see Eq. (117) in the Appendix A) as well as the following identity (see Eq. (122) in the Appendix A):

\[
\mathbf{F}^{T} \mathbf{m} \mathbf{F}^{\text{inv}} = \mathbf{M} \mathbf{C}^{\text{inv}},
\]

and conclude:

\[
\mathbf{F}^{T} \mathbf{m} \mathbf{F}^{\text{inv}} = \mathbf{M} \mathbf{C}^{\text{inv}}.
\]

We observe that tensor \( \mathbf{M} \) is unsymmetric. In order to obtain a constitutive model in which all quantities are symmetric, we follow the recently proposed approach in Arghavani et al. (2010a) and utilize the following identity (see Eq. (124) in the Appendix A):

\[
\log(\mathbf{C}^{\text{inv}}) = \mathbf{U}^{\text{inv}} \log(\mathbf{U}^{\text{inv}}) \mathbf{C}^{\text{inv}},
\]

and express tensor \( \mathbf{M} \) as

\[
\mathbf{M} = \mathbf{U}^{\text{inv}} \mathbf{Q} \mathbf{U}^{\text{inv}}^{-1},
\]

where the symmetric tensor \( \mathbf{Q} \) is expressed as

\[
\mathbf{Q} = \alpha_1 \mathbf{1} + 2 \alpha_2 \mathbf{H}^e + 4 \alpha_3 \mathbf{H}^{2}.
\]

Comparing (48) and (40), and considering (41), we use the representation theorem and express (48) as

\[
\mathbf{Q} = \frac{\partial \psi^e(\mathcal{H})}{\partial \mathcal{H}}.
\]

We now substitute (47) into (45), which yields:

\[
\mathbf{F}^{T} \mathbf{m} \mathbf{F}^{\text{inv}} = \mathbf{U}^{\text{inv}} \mathbf{Q} \mathbf{U}^{\text{inv}}.
\]

Using the isotropic property of \( \psi^{\text{inv}} \), following a similar approach, we obtain:

\[
\mathbf{x} = \frac{\partial \psi^{\text{inv}}}{\partial \mathbf{h}^{\text{inv}}} = \beta_1 \mathbf{1} + \beta_2 \mathbf{h}^{\text{inv}} + \beta_3 \mathbf{h}^{2},
\]

where

\[
\beta_i = \beta_i(I_{1e}, II_{1e}, III_{1e}, T) = \beta_i(I_{1e}, II_{1e}, III_{1e}, T).
\]

\(^{10} \) Representation theorem: A tensor function \( \mathbf{D} = f(\mathbf{A}) \) is isotropic if and only if it has a representation of the form \( \mathbf{D} = f(\mathbf{A}) = \phi_0 \mathbf{1} + \phi_1 \mathbf{A} + \phi_2 \mathbf{A}^2 + \cdots + \phi_n \mathbf{A}^n \) where the \( \phi_i \) are invariants of \( \mathbf{A} \) and hence can be expressed as functions \( \phi_i = \phi_i(I_1, \ldots, I_n) \) of the principal invariants of \( \mathbf{A} \) while \( n \) denotes the order of tensor \( \mathbf{A} \). In the case when \( n = 3 \) the representation theorem reads \( \mathbf{D} = f(A) = \phi_0 \mathbf{1} + \phi_1 \mathbf{A} + \phi_2 \mathbf{A}^2 \) (Truesdell and Noll, 1965; Ogden, 1984; Ottosen and Ristinmaa, 2005).
We now use the identity $F^a_i h^a_m F^m_n = H^n c^n$ to obtain:

$$F^a_i x F^m_n = X c^n,$$  \hspace{1cm} (53)

where

$$X = \beta_1 1 + \beta_2 H^m + \beta_3 H^n.$$  \hspace{1cm} (54)

Similarly, comparing (51) and (54) and considering (52), we can express (54) as

$$X = \psi^{\infty}(H^m).$$  \hspace{1cm} (55)

Since $X$ and $U^m$ are coaxial, the following identity holds:

$$X = U^m X U^m.$$  \hspace{1cm} (56)

Substituting (56) into (53), we obtain:

$$F^a_i x F^m_n = U^m X U^m.$$  \hspace{1cm} (57)

Combining (50) and (57), we finally conclude:

$$F^a_i x F^m_n = F^a_i (m - x) F^m_n = U^m Z U^m,$$  \hspace{1cm} (58)

where

$$Z = Q - X.$$  \hspace{1cm} (59)

Using (58), we obtain:

$$F^a_i x F^m_n = U^m Z^m U^m \text{ and } tr(x^n) = tr(Z^n),$$  \hspace{1cm} (60)

where $n$ is a non-negative integer number. Eq. (60) means that the invariants of $x$ and $Z$ are the same.

We now assume $G(a, q)$ and $F(a, q)$ to be isotropic functions of $a$ which means that they are expressed only in terms of the invariants of $a$ which are equal to those of $Z$. Therefore, we present the Lagrangian form of the potential and limit functions, respectively, as follows:

$$G = G(Z, q) \quad \text{and} \quad F = f(Z, q) - R.$$  \hspace{1cm} (61)

In order to obtain the evolution equation in a Lagrangian form, we use relation $\dot{c}^m = 2F^a_i \dot{a}^m F^m_n$ and (35) to obtain:

$$\dot{c}^m = 2 \lambda F^a_i \frac{\partial G(a, q)}{\partial a} F^m_n.$$  \hspace{1cm} (62)

Considering the isotropic property of $G(x, q)$, the following identity holds (see Eq. (128) in the Appendix A):

$$F^a_i \frac{\partial G(a, q)}{\partial a} F^m_n = U^m \frac{\partial G(Z, q)}{\partial Z} U^m.$$  \hspace{1cm} (63)

Finally, substituting (63) into (62), we obtain the Lagrangian form of the tensorial evolution equation as follows:

$$\dot{c}^m = 2 \lambda U^m \frac{\partial G(Z, q)}{\partial Z} U^m.$$  \hspace{1cm} (64)

Moreover, the scalar internal variable evolution is stated as

$$\dot{j}^m = \lambda \frac{\partial G(Z, q)}{\partial q}.$$  \hspace{1cm} (65)

The second Piola–Kirchhoff stress tensor $S$ is obtained from the Kirchhoff stress tensor as

$$S = F^{-1} \tau F^{-T}.$$  \hspace{1cm} (66)

Substituting (17) and $\tau = s + p 1$ into (66) yields:

$$S = J^{-2/3} \left( p \bar{c}^{-1} + F^{-1} s F^{-T} \right).$$  \hspace{1cm} (67)

Substituting (30) and (43) into (67), we obtain:

$$S = J^{-2/3} \left( \dot{\phi} \bar{c}^{-1} + \bar{c}^{-1} M \right),$$  \hspace{1cm} (68)

where, $\dot{\phi} = \partial \phi / \partial q^r$. 
We now substitute (47) into (68) and obtain the second Piola–Kirchhoff stress tensor as follows\(^\dagger\):

\[
S = J^{-2/3} \mathbf{C}^{-1} \left( \mathbf{a} + \mathbf{U}^m \mathbf{Q} \mathbf{U}^{-1} \right).
\]

We finally summarizes the time-continuous finite-strain constitutive model, written only in terms of Lagrangian quantities in Table 1.

It is interesting that the proposed finite strain and the corresponding small strain (when the proposed model is linearized) constitutive models have similar forms, thanks to the Hencky strain. In particular, when linearized, the elastic-like strain tensor \(\mathbf{H}\) is reduced to small-strain elastic strain tensor, i.e.

\[
\mathbf{H}^e \approx \varepsilon - \dot{\varepsilon}^m + \text{higher order terms} \approx \varepsilon^e,
\]

where \(\varepsilon, \varepsilon^e\) and \(\dot{\varepsilon}^m\) are, respectively, the total, elastic and inelastic small strain tensors. Eq. (70) makes clear the reason for choosing the name “elastic-like strain tensor” for \(\mathbf{H}^e\).

4. The proposed constitutive model in the time-discrete frame

In this section, we investigate the time integration of the constitutive model derived in Section 3 with the final goal of using it within a finite element program. The main task is to apply an appropriate numerical time integration scheme to the evolution equation of the internal variables. In general, implicit schemes are preferred because of their stability at larger time step sizes.

We treat the nonlinear problem described in Section 3 as an implicit time-discrete deformation-driven problem. Accordingly, we subdivide the time interval \([0, T]\) in sub-increments and we solve the evolution problem over the generic interval \([t_n, t_{n+1}]\) with \(t_{n+1} > t_n\). To simplify the notation, we indicate with the subscript \(n\) a quantity evaluated at time \(t_n\), and with no subscript a quantity evaluated at time \(t_{n+1}\).

Assuming to know the solution and the deformation gradient \(F\) at time \(t_n\), as well as the deformation gradient \(\dot{F}\) at time \(t_{n+1}\), the internal variables (and the stress tensor) should be updated from the deformation history.

Since, the constitutive model is in terms of Lagrangian quantities, from now on, we consider as deformation driver the right Cauchy–Green deformation tensor (instead of the deformation gradient), assuming \(\mathbf{C}\) and \(\theta\) are given (control variables).

4.1. Time integration

We use a backward-Euler integration scheme for the scalar internal variable evolution Eq. (65) and obtain:

\[
\dot{\theta}^m = \dot{\theta}_0^m + \Delta \lambda \frac{\partial G(\mathbf{Z}, \mathbf{q})}{\partial \mathbf{q}},
\]

where \(\Delta \lambda = \lambda - \dot{\lambda}^m\).

However, time integration of the tensorial internal variable \(\mathbf{C}^m\) is not straightforward and the time-discrete form should satisfy the incompressibility constraint \(\det(\mathbf{C}^m) = 1\).

To this end, exponential-based integration schemes are frequently applied to problems in plasticity and isotropic inelasticity (Eterovic and Bathe, 1990; Miehe, 1996). The use of the exponential mapping enables to exactly conserve the inelastic volume. Thus, it allows larger time step sizes than any other first-order accurate integration scheme.

We now consider the following form of the evolution equation and investigate its time-discrete form:

\[
\dot{\mathbf{C}}^m = \lambda \mathbf{A} \quad \text{and} \quad \det(\mathbf{C}^m) = 1.
\]

Applying the exponential mapping scheme to the evolution Eq. (72), Reese and Christ (2008) have proposed the following time-discrete form which has also been used in several works (Christ and Reese, 2009; Evangelista et al., 2009; Vladimirov et al., 2008, 2010; Arghavani et al., 2010b):

\[
-\mathbf{C}^{-1} \mathbf{U}^{-1} \exp \left( \Delta \lambda \mathbf{U}^{-1} \mathbf{A} \mathbf{U}^{-1} \right) \mathbf{U}^{-1} = \mathbf{0}.
\]

Recently Arghavani et al. (2010b) have proposed an alternative form of the time-discrete evolution Eq. (73) through the following logarithmic mapping which is more robust and efficient when compared with (73):

\[
-\mathbf{U} \log \left( \mathbf{U} \mathbf{C}^{-1} \mathbf{U}^m \right) \mathbf{U}^m + \Delta \lambda \mathbf{A} = \mathbf{0}.
\]

In Table 2, we have summarized the time-integration algorithms (Arghavani et al., 2010b).

\(^\dagger\) According to Eq. (69), the so-called Mandel stress is \(\mathbf{CS} = \mathbf{a} + \mathbf{U}^m \mathbf{Q} \mathbf{U}^{-1} = \mathbf{U}^m (\mathbf{a} + \mathbf{Q}) \mathbf{U}^{-1}\) which is, in general, an asymmetric tensor.
Solution of system of Eq.(79) is, in general, approached through a straightforward Newton–Raphson method to find \( \phi^m \) and \( \theta^m \).

We now particularize the inelastic model and the corresponding solution algorithm presented in Sections 3 and 4 to a particular class of materials known as shape memory alloys.

Intelligent, smart or functional materials exhibit special properties that make them a suitable choice for industrial applications in many branches of engineering. Among different types of smart materials, shape memory alloys (SMAs) have unique features known as pseudo-elasticity, one-way and two-way shape memory effects (Otsuka and Wayman, 1998; Duerig et al., 1990). The origin of SMA material features is a reversible thermo-elastic martensitic phase transformation between a high symmetry, austenitic phase and a low symmetry, martensitic phase.

When the transformation is driven by a temperature decrease, martensite variants compensate each other, resulting in no macroscopic deformation. However, when the transformation is driven by the application of a load, specific martensite variants, favorable to the applied stress, are preferentially formed, exhibiting a macroscopic shape change in the direction of the...
applied stress. Upon unloading or heating, this shape change disappears through the reversible conversion of the martensite variants into the parent phase (Otsuka and Wayman, 1998; Funakubo, 1987).

During the forward transformation, under zero load, austenite begins to transform to twinned martensite at the martensitic start temperature ($M_s$) and completes transformation to martensite at the martensitic finish temperature ($M_f$). Similarly, during heating, the reverse transformation initiates at the austenitic start temperature ($A_s$) and the transformation is completed at the austenitic finish temperature ($A_f$). Applying a stress at a temperature above $A_y$, SMAs exhibit pseudo-elastic behavior with a full recovery of inelastic strain upon unloading, while at a temperature below $M_s$, the material presents the shape memory effect with permanent inelastic strains upon unloading which may be recovered by heating.

In most applications, SMAs experience complicated multi-axial loadings, typically undergoing very large rotations and moderate strains (i.e., in the range of 10–15% for polycrystals Shaw, 2000). For example, with reference to biomedical applications, stent structures are usually designed to significantly reduce their diameter during the insertion into a catheter; thereby, large rotations combined with moderate strains occur.

The so-called variant reorientation has been observed to be the main phenomenon in non-proportional loadings of SMAs (Lim and McDowell, 1999; Sittner et al., 1995; Helm, 2001, 2002; Bouvet et al., 2002; Grabe and Bruhns, 2009). Recently, Grabe and Bruhns (2009) have conducted several multi-axial experiments on polycrystalline NiTi within a wide temperature range, showing the strong nonlinearity as well as the path dependencies of the response and highlighting the presence of reorientation processes for complex loading paths.

These experimental observations, as well as situations experienced by SMAs in real engineering applications, call for the development of a 3D SMA constitutive model, taking into account both finite strain and variant reorientation under general loading conditions (multi-axial non-proportional loadings).

Up to now several 3D SMA constitutive models have been proposed in the small deformation regime (Raniecki et al., 1992, 1994; Leclercq and Lexcellent, 1996; Bekker and Brinson, 1997; Raniecki and Lexcellent, 1998; Souza et al., 1998; Qidwai and Lagoudas, 2000; Helm and Haupt, 2003; Auricchio et al., 2003; Liu et al., 2004; Lagoudas and Entchev, 2004; Auricchio et al., 2007; Popov and Lagoudas, 2007; Panico and Brinson, 2007; Thiebaud et al., 2007; Moumni et al., 2008; Arghavani et al., 2010c).

Finite strain SMA constitutive models available in the literature have been mainly developed by extending small strain constitutive models. The approach in most of the cases is based on the multiplicative decomposition of the deformation gradient into an elastic and an inelastic or transformation part (Auricchio and Taylor, 1997; Helm, 2001; Auricchio, 2001; Pethö, 2001; Ziolkowski, 2007; Helm, 2007; Reese and Christ, 2008; Christ and Reese, 2009; Evangelista et al., 2009; Thamburaja, 2010; Arghavani et al., 2010d), although there are some models in the literature which have utilized an additive decomposition of the strain rate tensor into an elastic and an inelastic part (Müller and Bruhns, 2004, 2006). Moreover, some finite strain non-phenomenological models have been proposed in the literature (see e.g., Patoor et al., 1996; Thamburaja and Anand, 2001, 2005; Pan et al., 2007a,b; Stein and Sagar, 2008 among others).

In the constitutive models based on the multiplicative decomposition of the deformation gradient, Green–Lagrange strain has been used as a strain measure and in most cases the obtained constitutive models are in a Lagrangian form. In the approaches based on the use of Green–Lagrange strain, one critical point is introducing a relation between martensite volume fraction, $\frac{\varepsilon}{3}$, and the norm of the Green–Lagrange strain measure $e_L$ (since $0 < \frac{\varepsilon}{3} < 1$ it can be considered as a constraint on the inelastic strain). While the martensite fraction is a physical quantity, the Green–Lagrange strain is not a physical measure of strain. Considering this point, the proposed finite strain model which is based on the true strain is preferred.

On the other hand, in the constitutive models in which logarithmic strain is used as a (physical) strain measure, the additive decomposition of the strain rate tensor is adopted and the final form of the constitutive equations are Eulerian and objective rates should be used. For example, Müller and Bruhns (2004, 2006) have used a corotational rate associated to the logarithmic spin in their formulations.

In this work, we used the logarithmic strain as a physical measure of strain and based on the multiplicative decomposition of the deformation gradient we present a constitutive model in a Lagrangian form. Moreover, we develop a finite-strain SMA model by extending a small-strain constitutive model proposed by Souza et al. (1998), its capability has been shown in several works (Auricchio and Petrini, 2002, 2004a,b; Arghavani et al., 2010c). The small-strain model is capable in predicting both super-elasticity and shape memory effect as well as variant reorientation under non-proportional loading; it also has a suitable form which makes it possible to develop a robust integration algorithm as well as a successful computational tool (Arghavani et al., 2010b).

To this end, we use the proposed constitutive model in Section 3 (summarized in Table 1) and assuming appropriate form for strain energy functions in Section 5.1, we propose a Hencky-based finite strain constitutive model for shape memory alloys. In Section 5.2, we present a non-singular, continuous definition for the model variables, avoiding extensively-used regularization schemes. Section 5.3 describes the solution algorithm followed by defining a Nucleation–Completion condition (Arghavani et al., 2010b) in Section 5.4 and consistent tangent construction in Section 5.5. Finally in Section 6, implementing the proposed SMA constitutive model as well as the solution algorithm into a user defined subroutine UMAT in the nonlinear finite element software ABAQUS/Standard, we solve several boundary value problems.

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12 It has been defined in several works (Arghavani et al., 2010d; Reese and Christ, 2008; Christ and Reese, 2009; Helm and Haupt, 2003; Evangelista et al., 2009) as $\frac{\varepsilon}{3} = \frac{z}{1 - z}$, where $e_L$ is a material parameter related to the maximum transformation strain.
5.1. Strain energy function forms

We now assume the following forms for the elastic strain energy functions\(^\text{13}\):

\[
\phi^e(\theta^e) = \frac{1}{2} K \theta^e, \quad \psi^e(h^e) = \mu \|h^e\|^2,
\]

(80)

where \(K\) and \(\mu\) are the material parameters.

Since experimental evidence indicates that phase transformation-induced deformations in SMAs are nearly isochoric, we have to impose \(\det(F^n) = 1\), which after taking the time derivative results in:

\[
\text{tr}(d^n) = 0.
\]

(81)

Therefore, \(f^n = 1\) (or \(\theta^n = 0\)) is a known quantity. For this reason, without loss of generality, we can assume \(\phi^{in}(\theta^n) = 0\). In the following, for simplicity, we denote all inelastic quantities without a bar symbol.

In order to describe SMAs behavior, we use the following form for \(\psi^{in}\) and we denote it by \(\psi^l\) to clarify it as the transformation strain energy function (Souza et al., 1998; Arghavani et al., 2010b):

\[
\psi^{in} = \psi_l(h^l, T) = \tau_M \|h^l\| + \frac{1}{2} h_k \|h^l\|^2 + I_{\xi_k} (\|h^l\|).
\]

(82)

where \(\tau_M = b (T - T_0)\) and \(b, T_0\) as well as the kinematic hardening parameter \(h_k\) are material parameters. Also, the Macaulay brackets calculate the positive part of the argument, i.e. \((x) = (x + |x|)/2\) and the norm operator is defined as \(\|A\| = \sqrt{A : A}\).

Moreover, in Eq. (82) we use the indicator function \(I_{\xi_k}\) defined as

\[
I_{\xi_k} (\|h^l\|) = \begin{cases} 
0 & \text{if } \|h^l\| \leq \xi_k, \\
+\infty & \text{otherwise},
\end{cases}
\]

(83)

to satisfy the constraint on the transformation strain norm (i.e., \(\|h^l\| \leq \xi_k\)). The material parameter \(\xi_k\) is the maximum transformation strain norm in a uni-axial test.

We assume an associated case (\(F = G\)) and consider a von-Mises type limit (yield) function as\(^\text{14}\)

\[
F(Z, q) = \|Z\| - R.
\]

(84)

We now compute the strain energy functions derivative as follows:

\[
\phi^e = \frac{\partial \phi^e}{\partial \theta^e} = K \theta^e, \quad \psi^e = \frac{\partial \psi^e}{\partial h^e} = 2 \mu h^e,
\]

(85)

which yields:

\[
\alpha_1 = 0, \quad \alpha_2 = \mu, \quad \alpha_3 = 0.
\]

(86)

and accordingly:

\[
Q = \mu \log (U^{in} C U^{in})^{-1} = 2 \mu h^e.
\]

(87)

In a similar approach, we obtain:

\[
\frac{\partial \psi^l}{\partial h^l} = h_k h^l + (\tau_M(T) + \gamma) \frac{h^l}{\|h^l\|}.
\]

(88)

The positive variable \(\gamma\) results from the indicator function subdifferential \(\partial I_{\xi_k}(h^l)\) and it is defined as

\[
\begin{cases} 
\gamma \geq 0 & \text{if } \|h^l\| = \xi_k, \\
\gamma = 0 & \text{if } \|h^l\| < \xi_k.
\end{cases}
\]

(89)

We then conclude:

\[
\beta_1 = 0, \quad \beta_2 = h_k + \frac{\tau_M(T) + \gamma}{\|h^l\|}, \quad \beta_3 = 0.
\]

(90)

---

\(^{13}\) This is a Hookean-type strain energy function in terms of Hencky strain. It is also known as Hencky model in the literature (see e.g., Xiao and Chen (2002)).

\(^{14}\) For a kinematic hardening plasticity, we can assume a quadratic form for \(\psi^{in}\), denoted by \(\psi^p\):

\[
\psi^p = \psi^p(h^l, T) = \frac{1}{2} h_k \|h^l\|^2.
\]

Comparing the above equation and (82), we can conclude that a SMA constitutive model is reduced to a kinematic hardening plasticity model, when \(\tau_M(T) = 0\) (or \(T < T_0\)) and \(\xi_k \to \infty\). To this reason, The proposed SMA constitutive model is capable to describe associated von-Mises type kinematic hardening plasticity.
for the case of vanishing transformation strain. This is an issue in most of the SMA constitutive models (Evangelista et al., 2009; Helm and Haupt, 2003; Reese and Christ, 2008; Panico and Brinson, 2007) and a regularization scheme is extensively used in the literature to overcome this problem. This issue has recently been investigated by Arghavani et al. (2010c) and Arghavani et al. (2010b) in the small-strain and the finite-strain regime, respectively. They revised N as a well-defined, non-singular and continuous variable. Moreover, in Arghavani et al. (2010b) it is shown that using a regularization scheme increases solution time and consequently decreases computational efficiency. In fact, in a regularized scheme, most of the elastic behavior of the austenitic phase is considered as a phase transforming case for which a nonlinear system has to be solved. This can have a remarkable effect on the solution time for structures like stents in which considerable part of the structure remains elastic.

In the following, motivated by the work by Arghavani et al. (2010b), we suggest to avoid a regularized form for N and to deal with the case of vanishing transformation strain through a careful analytical study of the limiting conditions. We start investigating a condition in which \(H^f\) starts to evolve from a zero value (indicated in the following as \(\text{Nucleation}\)), i.e., \(H^f = 0\) with \(||H^f|| > 0\). Substituting \(U^f = 1\) into (59)\(^\text{15}\) we conclude that \(Z = Q_c\). We now consider the evolution equation which yields:

\[
\dot{\mathbf{C}} = 2\mu \frac{Q_c}{||Q_c||},
\]

where \(Q_c = \mu \log(\mathbf{C}) = 2\mu \mathbf{H}\) is obtained from (87) substituting 1 in place of \(U^f\). According to (92), the transformation strain, \(H^f = \frac{1}{2} \log(\mathbf{C}) = \frac{1}{2}(\mathbf{C} - 1), \) nucleates in the \(Q_c\) direction.

We now investigate the case when transformation strain vanishes from a nonzero value (indicated in the following as \(\text{Completion}\)), i.e., \(H^f = 0\) \((U^f = 1)\) with \(||H^f|| < 0\). Since \(||H^f|| = 0\), we conclude that adopting any arbitrary direction \(N\) leads to \(H^f = 0\). As stated in Arghavani et al. (2010c) (for a small-strain case), we select this direction as the \(Q_c\) direction which also guarantees continuity. Therefore, we revise the variable \(N\) in the following form:

\[
N = \begin{cases} 
\frac{Q_c}{||Q_c||} & \text{if } ||H^f|| = 0, \\
\frac{H^f}{||H^f||} & \text{if } ||H^f|| \neq 0.
\end{cases}
\]

We note that according to the above discussion, the tensor \(N\) and consequently the tensor \(X\) are well-defined, non-singular and continuous.

We finally summarize the proposed finite strain constitutive model for shape memory alloys in Table 3. We recall that, for simplicity, all inelastic quantities are denoted without a bar symbol.

5.3. Solution algorithm

As usual in computational inelasticity problems, to solve the time-discrete constitutive model we use an elastic predictor-inelastic corrector procedure. The algorithm consists of evaluating an elastic trial state, in which the internal variable remains constant, and of verifying the admissibility of the trial function. If the trial state is admissible, the step is elastic; otherwise, the step is inelastic and the transformation internal variable has to be updated through integration of the evolution equation.

We now substitute (84) into (78) and obtain the time-discrete tensorial evolution equation as

\[
-\log \left( \mathbf{U}^f \mathbf{C}_n^{\text{trans}} \mathbf{U}^f \right) + 2\Delta \gamma \frac{Z}{||Z||} = 0.
\]

In order to solve the inelastic step, we use another predictor–corrector scheme, that is, we assume \(\gamma = 0\) (i.e., we predict an unsaturated transformation strain case with \(||H^f|| \leq e_L\)) and we solve the following system of nonlinear equations (we refer to this system as \(\text{PT1}\) system):

\[
\begin{cases}
R^t = -\log \left( \mathbf{U}^f \mathbf{C}_n^{\text{trans}} \mathbf{U}^f \right) + 2\Delta \gamma \mathbf{W} = 0, \\
R^t = ||Z|| - R = 0,
\end{cases}
\]

where \(\mathbf{W} = Z/||Z||\). If the solution is not admissible (i.e., \(||H^f||_{\text{PT1}} > e_L\), we assume \(\gamma > 0\) (i.e., we consider a saturated transformation case with \(||H^f|| = e_L\)) and we solve the following system of nonlinear equations (we refer to this system as \(\text{PT2}\) system):

\(^{15}\) To avoid confusing, we remark that we have used superscript "\(t\)" in place of "\(in\)" in this section.
5.4. Considerations on nucleation-completion condition

We first investigate the trial value of the limit function in the nucleation case as

\[
F_{TR} = ||Q_n|| - \tau_M(T) - R = ||Q_n|| - \tau_M(T) - R > 0, \tag{97}
\]

where a subscript TR indicates a trial value. We may thus introduce the following nucleation condition:

\[
||Q_n|| > \tau_M(T) + R \text{ and } ||H_n^e|| = 0. \tag{98}
\]

In the solution procedure, when ||H_n^e|| = 0 we check the nucleation condition and if not satisfied, we assume an elastic behavior.

We remark that, to avoid singularity in system (95) for the nucleation case, it is necessary to use a nonzero initial (guess) transformation strain. Since, we know the initial transformation strain direction from (93), the only unknown for constructing an initial guess is its norm; if we denote with \(\tilde{q}\) its value, we can use the following initial guess for the nucleation case:

\[
H_0^e = \log(U_0^e) \approx U_0^e - 1 = \tilde{q}N \Rightarrow U_0^e = 1 + \tilde{q}N, \tag{99}
\]

where a subscript 0 denotes the initial guess. A value of 10^{-4} could be an appropriate choice for \(\tilde{q}\).

We now focus on the completion case, i.e., \(H_n^e \neq 0\) but \(H^e = 0\). To this end, we consider Eqs. (76) or (77) and substitute \(U^e = 1\) (as it happens for the completion case) to obtain:

\[
\Delta \bar{A} = \log \left(C_n^{-1}\right) = -2H_n^e; \quad A = 2 \frac{Z}{||Z||}; \quad Z = Q_n - \tau_M N. \tag{100}
\]

To this end, using (100)_{1,2} which define the \(Z\) direction and considering the limit function (84), we conclude:

\[
Z = -R \frac{H_n^e}{||H_n^e||} = -RN_{TR}, \tag{101}
\]

substituting (101) in (100)_{1}, we obtain:

\[
Q_n + RN_{TR} = \tau_M N. \tag{102}
\]

Finally taking the norm of both sides of Eq. (102), we define the following completion condition:

\[
||Q_n + RN_{TR}|| \leq \tau_M; \quad ||H_n^e|| \neq 0. \tag{103}
\]

Therefore, in the solution procedure we also should check the completion condition and, if it is satisfied, we simply update the internal variable by setting \(U^e = 1\).
Table 4 finally presents the branch detection and the solution algorithm. For more details and discussions on the algorithm robustness, we refer to Arghavani et al. (2010b).

5.5. Consistent tangent matrix

In the following, we linearize the nonlinear equations as it is required for the iterative Newton–Raphson method. For brevity, we report the construction of the tangent matrix only for the case of the saturated phase transformation, corresponding to (96). Linearizing (96), we obtain:

\[
\begin{align*}
R^c + R^c_{\lambda} : dU^c + R^c_{\lambda} : d\Delta\lambda + R^c_{\gamma} : d\gamma &= 0, \\
R^f + R^f_{\lambda} : dU^f + R^f_{\lambda} : d\Delta\lambda + R^f_{\gamma} : d\gamma &= 0, \\
R^y + R^y_{\lambda} : dU^y + R^y_{\lambda} : d\Delta\lambda + R^y_{\gamma} : d\gamma &= 0,
\end{align*}
\]

(104)

where subscripts following a comma indicate differentiation with respect to that quantity. The derivatives appearing in the above equation are detailed in Arghavani (2010).

Utilizing the linearized form (104), after converting it to a matrix form, a system of eight nonlinear scalar equations is solved to obtain \( dU^c, d\Delta\lambda \) and \( d\gamma \).

We now address the construction of the tangent tensor consistent with the time-discrete constitutive model. The use of a consistent tensor preserves the quadratic convergence of the Newton–Raphson method for the incremental solution of the global time-discrete problem, as in the framework of a finite element scheme.

The consistent tangent is computed by linearizing the second Piola-Kirchhoff tensor, i.e.

\[
d\mathbf{S} = \mathbf{D} : d\mathbf{E} = \frac{1}{2} \mathbf{D} : d\mathbf{C}.
\]

(105)

Recalling that \( \mathbf{S} \) is a function of \( \mathbf{J}, \mathbf{C} \) and \( \mathbf{U}^i \), we can write:

\[
d\mathbf{S} = \frac{\partial \mathbf{S}}{\partial \mathbf{J}} d\mathbf{J} + \frac{\partial \mathbf{S}}{\partial \mathbf{C}} d\mathbf{C} + \frac{\partial \mathbf{S}}{\partial \mathbf{U}^i} d\mathbf{U}^i.
\]

(106)

The following relations express \( d\mathbf{J} \) and \( d\mathbf{C} \) in terms of \( d\mathbf{C} \) (Holzapfel, 2000):

\[
d\mathbf{J} = \frac{1}{2} \mathbf{C}^{-1} : d\mathbf{C} \quad \text{and} \quad d\mathbf{C} = \frac{1}{2} \mathbf{p} : d\mathbf{C},
\]

(107)

where,

\[
\mathbf{p} = \mathbb{I} - \frac{1}{3} \mathbf{C} \otimes \mathbf{C}^{-1},
\]

(108)

and \( \mathbb{I} \) denotes the fourth-order identity tensor.

<table>
<thead>
<tr>
<th>Table 4</th>
<th>Branch detection procedure and solution algorithm.</th>
</tr>
</thead>
<tbody>
<tr>
<td>1. Compute ( \mathbf{C} = \mathbf{P}^T \mathbf{F} ) then ( J = \sqrt{\det \mathbf{C}} ) and ( \mathbf{P} = J^{2/3} \mathbf{C} )</td>
<td></td>
</tr>
<tr>
<td>2. Compute ( \mathbf{Q}_p )</td>
<td></td>
</tr>
<tr>
<td>3. if ( (|\mathbf{Q}_p| &lt; 1 + \sigma + |\mathbf{H}_0| = 0) ) then</td>
<td>set ( \mathbf{U}^f = \mathbf{F}^T )</td>
</tr>
<tr>
<td>else</td>
<td>use (93) and set ( \mathbf{U}^f_{\text{TR}} = \mathbf{U}^f ) to compute ( f^T ) and trial solution</td>
</tr>
<tr>
<td>if ( (f^T &gt; 0) ) then</td>
<td>set trial solution as the problem solution, i.e., ( \mathbf{U}^f = \mathbf{U}^f_{\text{TR}} )</td>
</tr>
<tr>
<td>else if (completion condition (103)) then</td>
<td>set ( \mathbf{U}^f = \mathbf{U}^f_{\text{TR}} )</td>
</tr>
<tr>
<td>else</td>
<td>set ( \mathbf{U}^f_1 = \mathbf{U}^f ) and ( \Delta\lambda_0 = 0 )</td>
</tr>
<tr>
<td>if ( (|\mathbf{H}_0| = 0) ) then</td>
<td>set ( \mathbf{U}^f_1 ) value by (99)</td>
</tr>
<tr>
<td>solve PT1</td>
<td></td>
</tr>
<tr>
<td>if ( (|\mathbf{H}<em>0|</em>{\text{at}} &lt; \varepsilon) ) then</td>
<td>set PT1 solution as the solution: i.e., ( \mathbf{U}^f = \mathbf{U}^f_{\text{TR}} )</td>
</tr>
<tr>
<td>else</td>
<td>set ( \mathbf{U}^f_0 = \mathbf{U}^f_{\text{TR}} ), ( \Delta\lambda_0 = \Delta\lambda_{\text{PT1}} ) and ( \gamma_0 = 0 ), then solve PT2</td>
</tr>
<tr>
<td>set ( \mathbf{U}^f = \mathbf{U}^f_{\text{PT2}} )</td>
<td></td>
</tr>
<tr>
<td>end if</td>
<td></td>
</tr>
<tr>
<td>end if</td>
<td></td>
</tr>
<tr>
<td>end if</td>
<td></td>
</tr>
</tbody>
</table>
We now consider Eq. (96) as a function of $\mathbf{C}$, $\mathbf{U}^f$, $\Delta \lambda$ and $\gamma$, and then the corresponding linearization gives:

\[
\begin{bmatrix}
R_{\xi}^T : d\mathbf{C} + R_{\xi u}^T : d\mathbf{U}^f + R_{\xi \lambda}^T : d\Delta \lambda + R_{\xi \gamma}^T : d\gamma = 0, \\
R_{\eta}^T : d\mathbf{C} + R_{\eta u}^T : d\mathbf{U}^f + R_{\eta \lambda}^T : d\Delta \lambda + R_{\eta \gamma}^T : d\gamma = 0, \\
R_{\zeta}^T : d\mathbf{C} + R_{\zeta u}^T : d\mathbf{U}^f + R_{\zeta \lambda}^T : d\Delta \lambda + R_{\zeta \gamma}^T : d\gamma = 0.
\end{bmatrix}
\]  

Recasting (109) into a matrix form, we obtain:

\[
\begin{bmatrix}
d\mathbf{U}^f \\
d\Delta \lambda \\
d\gamma
\end{bmatrix} = - \begin{bmatrix}
R_{\xi u}^T & R_{\xi \lambda}^T & R_{\xi \gamma}^T \\
R_{\eta u}^T & R_{\eta \lambda}^T & R_{\eta \gamma}^T \\
R_{\zeta u}^T & R_{\zeta \lambda}^T & R_{\zeta \gamma}^T
\end{bmatrix}^{-1} \begin{bmatrix}
R_{\xi}^T \\
R_{\eta}^T \\
R_{\zeta}^T
\end{bmatrix} d\mathbf{C},
\]

where we have used $[\ ]$ to denote the matrix form of the tensorial argument. Now, using (110), we can compute the matrix $[B]$ (the matrix form of the fourth-order tensor $B$) such that:

\[
d\mathbf{U}^f = [B][d\mathbf{C}] \quad \text{or} \quad d\mathbf{U}^f = B : d\mathbf{C}.
\]

We then substitute (107) and (111)2 into (105) to obtain the consistent tangent as

\[
D = J^{1/3}J \otimes \mathbf{C}^{-1} + 2J^{-2/3}(S + T : B) : P,
\]

where the second-order tensor $\delta S/\delta J$ as well as the fourth-order tensors $\delta S/\delta \mathbf{C}$ and $\delta S/\delta \mathbf{U}^f$ are denoted by $J$, $S$ and $T$, respectively.

6. Numerical examples

In this section, we solve some boundary value problems to show the proposed model capabilities as well as the proposed integration algorithm and the solution procedure. A uni-axial test on a cube, crimping of a pseudo-elastic medical stent and a helical spring actuator are simulated to show the model capability of capturing both pseudo-elasticity and shape memory effect.

For all simulations, we use the commercial nonlinear finite element software ABAQUS/Standard, implementing the described algorithm within a user-defined subroutine UMAT.

The following material properties, typical of NiTi,\textsuperscript{16} are adopted in all the simulations of this paper:

\[
\begin{align*}
E &= 51700 \text{ MPa}, \quad v = 0.3, \quad h_k = 1000 \text{ MPa}, \quad \varepsilon_k = 10\%, \\
\beta &= 5.6 \text{ MPa } \text{ °C}^{-1}, \quad T_0 = -25 \text{ °C}, \quad R = 140 \text{ MPa}, \quad A_f = 0 \text{ °C}, \quad M_f = -25 \text{ °C}.
\end{align*}
\]

where $E$ and $v$ are the elastic modulus and the Poisson ratio, respectively.

6.1. Uni-axial test

We simulate a single element unit cube under an applied force on one face while the opposite face is fixed. The applied force is increased from zero to a maximum value of 1500 N and subsequently decreased to zero and increased in the opposite direction to a value of 1500 N and finally decreased back to zero. Fig. 1(a) and (b) show the SMA behavior at 37 and $-25 \text{ °C}$, respectively. We observe the pseudo-elastic behavior in Fig. 1(a) where the temperature is above the austenite finish temperature while in Fig. 1(b) the material does not recover the original shape after unloading; however it can be recovered by heating above the austenite finish temperature (shape memory effect).

6.2. Pseudo-elastic stent

In this example, the crimping of a pseudo-elastic medical stent is simulated at a temperature of 37 °C. To this end a medical stent with 0.216 mm thickness and an initial outer diameter of 6.3 mm (Fig. 2(a)) is cramped to an outer diameter of 1.5 mm. Utilizing the ABAQUS/Standard contact module, the contact between the catheter (with initial diameter of 6.5 mm) and the stent is considered in the simulation. A radial displacement of 2.5 mm is applied to the catheter and is then released to reach the initial diameter. In this process, the stent recovers its original shape after unloading. Fig. 2(b) shows the stent deformed shape when fully cramped. Moreover, Fig. 3 reports the von-Mises stress at the critical point $A$ (shown in Fig. 2(b)) versus the reduction percentage in catheter diameter.

\textsuperscript{16} See, e.g., www.shape-memory-alloys.com
Fig. 1. Illustration of the material behavior under uni-axial force-controlled test: (a) pseudo-elasticity at $T = 37 \, ^\circ\text{C}$, (b) shape memory effect at $T = -25 \, ^\circ\text{C}$.

Fig. 2. Pseudo-elastic stent crimping: (a) initial geometry, (b) crimped shape.

Fig. 3. von-Mises stress versus percentage of catheter diameter reduction (at point A shown in Fig. 2(b)).
6.3. Spring actuator

In order to investigate the shape memory effect, a vertical helical spring (with a wire diameter of 4 mm, a spring external diameter of 24 mm, a pitch size of 12 mm and with two coils and an initial length of 28 mm) is simulated using 9453 quadratic tetrahedron (C3D10) elements and 15764 nodes (Fig. 4A). The spring is loaded by a weight $W = 244$ N in martensite state at $T = -25 ^\circ C$ (Fig. 4B). If the stress applied by the weight is less than the alloy recovery stress, heating the material above $A_f$ induces the inverse phase transformation and the spring lifts the weight (Fig. 4C). Moreover, cooling below $M_f$ the transformation from austenite to martensite takes place and the weight stretches again the spring (Fig. 4D); accordingly, a repeatable two-way motion takes place.

We remark that in the analysis, we assume that during each time step the temperature is uniform in the material body, i.e., we neglect the thermo-mechanical coupling.

Fig. 5a shows the loading history during the simulation. In Fig. 5(b), we plot the vertical displacement versus temperature. It is observed that applying a temperature cycle between $-25$ and $75 ^\circ C$, the spring actuator traverse a length of 37 mm.

**Fig. 4.** Spring actuator: (A) initial geometry and mesh, (B) deformed shape due to the weight application at $T = -25 ^\circ C$, (C) spring shape recovery and weight lifting due to heating to $T = 75 ^\circ C$, (D) spring stretching due to cooling to $T = -25 ^\circ C$.

**Fig. 5.** (a) Loading history during simulation, (b) vertical displacement of the lower loaded end of spring versus temperature variation.
7. Summary

In this work, we propose a finite strain Hencky-based kinematic-hardening constitutive model. The derivation is based on the multiplicative decomposition of the deformation gradient into elastic and inelastic parts. Moreover, in the derivations, we use the isotropic property and the fact that the corotational rate of the Eulerian logarithmic strain associated with the so-called logarithmic spin is the strain rate tensor. The developed constitutive model has a Lagrangian form. Moreover, defining an elastic-like strain tensor, the developed finite strain constitutive model takes a structure similar to that of small strain model, thanks to Hencky strain. Introducing a logarithmic mapping, we also present the time-discrete constitutive model. We then apply the developed model to shape memory alloys (SMAs) and present a well-defined, non-singular constitutive model. We finally discuss solution algorithm and consistent matrix derivation and simulate several SMA boundary value problems, demonstrating the modeling as well as numerical implementation capabilities.

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Appendix A

In this appendix, we present details of the relations we used in the previous sections. The logarithm of a tensor $B$ can be expanded as follows:

$$\log B = (B - 1) - \frac{1}{2}(B - 1)^2 + \frac{1}{3}(B - 1)^3 - \frac{1}{4}(B - 1)^4 + \cdots,$$

which is absolutely convergent if $\|B - 1\| < 1$, but $B \neq 0$.\(^\dagger\)

We substitute $B'$ for $B$ in (113) to obtain:

$$\log B' = 2\log h' = (B' - 1) - \frac{1}{2}(B' - 1)^2 + \frac{1}{3}(B' - 1)^3 - \frac{1}{4}(B' - 1)^4 + \cdots.$$

We now show that $F'^t sF'^{-t}$ is a symmetric tensor. To this end we substitute (40) into (30)\(_2\) to obtain:

$$s = \frac{\partial \psi}{\partial h'} = \frac{\partial \psi}{\partial h'} = \frac{\partial \psi}{\partial h'} = \frac{\partial \psi}{\partial h'}.$$

We then observe that:

$$F'^t (B' - 1)^n F'^{-t} = (C - 1)^n,$$

where $n$ is a positive integer number. Substituting (116) into (114), we conclude:

$$F'^t h' F'^{-t} = H' \quad \text{and} \quad F'^t h'^t F'^{-t} = H'^t.$$

We now combine (117) and (115) to obtain:

$$F'^t sF'^{-t} = \frac{\partial \psi}{\partial h'} + 2x_2 H' + 4x_3 H'^t.$$

According to (118), we conclude that $F'^t sF'^{-t}$ is a symmetric tensor.

We now show that $h' = \frac{1}{2} F'^{-t} \log(C \in \mathbb{C}^{n \times n}) F'^t$. To this end, we observe that:

$$h' = F'^t F'^{-t} = \frac{1}{2} F'^{-t} \log(C \in \mathbb{C}^{n \times n}) F'^t.$$

According to (119), we conclude:

$$h' = \frac{1}{2} F'^{-t} \log(C \in \mathbb{C}^{n \times n}) F'^t.$$

We now substitute (120) into (114) and obtain:

$$h' = \frac{1}{2} F'^{-t} \log(C \in \mathbb{C}^{n \times n}) F'^t.$$

\(^\dagger\) Similar expansion can be considered if $\|B - 1\| > 1$, for example if $\|B - 1\| < 1$, we use: $\log B = \log(2)1 + (B - 21) - \frac{1}{2}(B - 21)^2 + \frac{1}{3}(B - 21)^3 - \frac{1}{4}(B - 21)^4 + \cdots$. 

Substituting (121) into (117) we also obtain:
\[
F^{in} \mathbf{H} F^{in} = \frac{1}{2} \log \left( \mathbf{C}^{in-1} \right) \mathbf{C}^{in}.
\] (122)

We now mention the following properties:
\[
\mathbf{C}^{in-1} \mathbf{C}^{in-1} = \mathbf{U}^{in} \left( \mathbf{U}^{in-1} \mathbf{C}^{in-1} \mathbf{U}^{in-1} \right) \mathbf{U}^{in-1},
\]
\[
\left( \mathbf{C}^{in-1} \mathbf{C}^{in-1} \right) = \mathbf{U}^{in} \left( \mathbf{U}^{in} \mathbf{C}^{in-1} \mathbf{U}^{in} \right) \mathbf{U}^{in-1},
\] (123)

Finally, substitution of (123) into logarithmic expansion (113) yields:
\[
\log(\mathbf{C}^{in-1}) = \mathbf{U}^{in} \log \left( \mathbf{U}^{in} \mathbf{C}^{in-1} \mathbf{U}^{in} \right) \mathbf{U}^{in-1}.
\] (124)

We assumed \( G(x, q) \) to be an isotropic function of its argument, therefore, considering representation theorem, we may write:
\[
\frac{\partial G(x, q)}{\partial \mathbf{x}} = c_1 I + c_2 \mathbf{x} + c_3 \mathbf{x}^2,
\] (125)
where the coefficients \( c_1, c_2 \) and \( c_3 \) are functions of the invariants of \( \mathbf{x} \) and \( q \). We now multiply (125) from left and right by \( F^{in} \) and \( \mathbf{F}^{in} \), respectively:
\[
F^{in} \frac{\partial G(x, q)}{\partial \mathbf{x}} \mathbf{F}^{in} = c_1 \mathbf{F}^{in} \mathbf{F}^{in} + c_2 \mathbf{F}^{in} \mathbf{x} \mathbf{F}^{in} + c_3 \mathbf{F}^{in} \mathbf{x}^2 \mathbf{F}^{in}.
\] (126)

We now use (60) in (126) to obtain:
\[
F^{in} \frac{\partial G(x, q)}{\partial \mathbf{x}} \mathbf{F}^{in} = c_1 \mathbf{U}^{in} + c_2 \mathbf{U}^{in} \mathbf{Z} \mathbf{U}^{in} + c_3 \mathbf{U}^{in} \mathbf{Z}^2 \mathbf{U}^{in} = \mathbf{U}^{in} \left( c_1 I + c_2 \mathbf{Z} + c_3 \mathbf{Z}^2 \right) \mathbf{U}^{in}.
\] (127)

In Eq. (60) we showed that the invariants of \( \mathbf{x} \) are equal to those of \( \mathbf{Z} \). Thus, we conclude that the terms inside the parenthesis in (127) represent \( \mathbf{U}^{in} \mathbf{Z} \mathbf{U}^{in} \mathbf{Z} \mathbf{U}^{in} \). Therefore, we can write:
\[
F^{in} \frac{\partial G(x, q)}{\partial \mathbf{x}} \mathbf{F}^{in} = \mathbf{U}^{in} \left( \mathbf{Z} \frac{\partial G(x, q)}{\partial \mathbf{Z}} \mathbf{U}^{in} \right) \mathbf{U}^{in}.
\] (128)

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